he Structure of the Seven Layer Barium Niobium Silicon Oxide, $a_3Nb_{21-x}Si_2O_{44}$ ($x \simeq 0.2$)

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One of the products of reaction between Nb₂O₅, Nb, and BaO in a 1.6:1:2.4 mole ratio at 1250°C in an evacuated silica capsule is the complex oxide Ba₃Nb_{21-x}Si₂O₄₄. The compound has an hexagonal unit cell of dimensions $a = 7.774 \pm 0.004$ Å, $c = 16.765 \pm 0.007$ Å, probable space group $P\overline{3}$, Z = 1. Its structure has been determined from 4484 independent reflections measured by counter methods. Refinement by Fourier and least-squares methods was carried out to a weighted R value of 9.8% (conventional R = 12.4%).

The structure consists of a close-stacking of seven close-packed layers in the sequence BaO₆, O₆ (oxygen layer with 1/7 of the oxygen missing), O₇, BaO₆, BaO₆, O₇, O₆. Between two BaO₆ layers there is one Nb in an octahedral site (population parameter $\simeq 0.8$); between BaO₆ and O₇ layers there are three Nb in octahedral sites; between BaO₆ and O₆ layers there are three Nb in sites coordinated by five O in a square pyramidal arrangement; and between O₆ and O₇ layers, there are three Nb in similar square pyramidal sites, one Nb in an octahedral site, and one Si in a tetrahedral site. The niobiums in square pyramidal sites form an octahedron about the oxygen vacancy of the O₆ layer and engage in metal-metal bonding, the Nb-Nb distances ranging from 2.78 to 2.85 Å.

 Nb_6O_{18} clusters can be identified in which the Nb_6 octahedron is surrounded by a cuboctahedron of oxygen atoms capped by oxygen atoms on the square faces. Alternatively, the oxygen array may be described as an octahedron of oxygens with oxygens also at the centers of the edges.

ntroduction

When Nb₂O₅, Nb, and BaO react in the presence of SiO₂, various complex oxides may be produced. The first such compound to be lescribed was Ba₃Nb₆Si₄O₂₆, which Shannon letermined to have a structure composed of strings of NbO₆ octahedra joined by cornerharing, both among each other and with Si₂O₇ groups (two SiO₄ tetrahedra sharing one vertex), with bariums occupying holes between the chains (1). The compound Ba₃Nb₇Si₂O_{25-x} ($x \simeq 0.5$) has a similar but somewhat more complex structure (2), in which some edge-sharing of NbO₆ octahedra occurs. A third oxide, Ba₃Nb_{21-x}Si₂O₄₄, however, has an entirely different structure which will be described herein.

In it Nb_6O_{18} clusters can be identified. Professor Ward has always shown an interest in such clusters, so we are particularly pleased to be able to report on a compound which not only represents a continuation of his work at the University of Connecticut, but which also has some features of special interest to him.

Experimental

The reactants Nb₂O₅ and Nb were obtained commercially. BaO was prepared by heating BaO₂ in vacuo to 1000°C for one hour. These three reactants were mixed in a 1.6:1:2.4 mole ratio and triturated under petroleum ether. The resulting powder was placed in a silica capsule which was evacuated and sealed, the silica serving as both container and reactant. The sample was held at 1250°C for one week. Upon removal from the furnace, the capsule was in a partially collapsed state, and on cooling it became clouded and crazed on the inside.

When the capsule was opened, black, highly reflective crystals were observed, both loose and

embedded in the walls, many of the former aggregated into clusters that appeared to have been partially melted. The loose non-aggregated crystals occurred in two distinct habits: thick chunks only occasionally exhibiting flat faces, and very thin hexagonal plates, often broken but usually having at least one 120° angle visible. The chunks were eventually determined to be $Ba_3Nb_7Si_2O_{25-x}$. It is, however, the investigation of the thin plates that is of interest here.

A single crystal in the shape of a roughly triangular plate about 0.3 mm across and 0.05 mm thick was selected for study. X-ray precession photographs showed hexagonal (trigonal) symmetry, Laue group $\overline{3}$. There were no systematic absences, so the probable space group is either P3 or $P\overline{3}$. Unit-cell dimensions of a = 7.78 Å, c = 16.75 Å were determined from the photographs. Due to the extremely small quantity of sample, measurement of its density was not feasible.

Data for structure determination were collected on a Picker Nuclear computer-controlled singlecrystal diffractometer, using MoK σ radiation and operating in the θ -2 θ scan mode. The base scan width was 3° in 2θ ; a dispersion factor of 0.692 was used; background was counted for 20 seconds on each side of the peak; three standard reflections were measured every 40 reflections. Data were collected in the following range: *h* and *k*, -3 to 10; *l*, -3 to 36; and 2 θ , 2° to 100°. In all, 6260 intensities were measured. The standards showed considerable variation with time, which was later shown to be due to instrument instability. Partial correction for this variability was made by dividing the data into groups during whose collection the standards remained reasonably constant and applying scale factors to bring the standards of different groups into agreement. Twenty scale groups were used, containing from 80 to 920 reflections and with scale factors ranging from 0.928 to 1.071. After scaling, the three standards had standard deviations of 1.6%, 6.0%, and 3.6%. The data were corrected for background and absorption, and reflections related by the symmetry of the Laue group were averaged. The absorption correction was made using Prewitt's program ACACA (3), with a few modifications to compensate for errors presumably due to diffractometer alignment problems. Averaging reduced the data set to 4484 independent reflections.

Of these, 1148 had intensities less than twice the standard deviation of the background and were labelled as unobserved. Upper bounds for these intensities were set equal to the observe intensities plus twice their standard deviation $(I_0 + 2\sigma_{I_0})$. A standard deviation was calculate for each reflection according to the formula

$$\sigma_F = \frac{\sigma_g}{10} \frac{1}{2\sqrt{Lp}} \left(\{N_T + BG + [0.02(N_T - BG)]^2\} / (N_T - BG) \right)^1 \right)$$

for observed reflections and

$$\sigma_F = \frac{\sigma_g}{10} \frac{1}{2\sqrt{\mathrm{Lp}}} (\mathrm{BG})^{1/2}$$

for unobserved reflections, where σ_g is the percenstandard deviation in a group of symmetry related reflections (set to 10 for single reflections) Lp is the product of Lorentz and polarization factors, N_T is the total counts recorded on the scan through the peak, and *BG* is the background counts for the scan. These standard deviation are derived from counting statistics, the equation used being based on those given by Stout and Jensen (4).

Lorentz and polarization corrections were made using the program package "The X-ray System" (5), which was also used to calculate accurate cell dimensions of $a = 7.774 \pm 0.004$ Å and $c = 16.765 \pm 0.007$ Å from the 20 values o twelve individually centered reflections, and in all subsequent calculations of Fourier map. least squares refinements, and so forth.

A rough determination of relative compositior was made with a KEVEX-Quanta/Metrix energy dispersive spectrometer, using Ba_3Nb_7 Si_2O_{25-x} as the standard. This measuremen indicated a Ba/Si ratio of about 3/2 and a Ba/N!! ratio of about 1/7. The statistical distributior of normalized structure factors indicated the presence of a center of symmetry, making the probable space group $P\overline{3}$.

Structure Determination and Refinement

Examination of the hk0 reflections revealed 140 and 630 to be very strong, suggesting tha many of the heavy atoms lie on or near the inter sections of the corresponding planes, i.e. a sites 0, 0, z; 1/3, 2/3, z; 1/21, 5/21, z; 2/7, 3/7, z 2/21, 10/21, z (and symmetry-related sites) Of these, only the first two are suitable for barium, since use of any of the other three would place symmetry-related bariums uncomfortably close together. All five sites are possible for obium (and also for oxygen). A Patterson map as calculated and showed strong concentrations f peaks corresponding to the interatomic vectors nong the above hypothesized locations.

One particularly large peak indicated the posble presence of a heavy atom at 1/3, 2/3, 0.14. difference Fourier map calculated from a model f just one barium at this location was used in onjunction with the Patterson to suggest nioium locations. Structure solution proceeded y trial and error, using ΔF maps as the primary ool for revision of the model. In order to reduce alculation time, only observed reflections of $\sin\theta \le 0.5$ (1770 reflections) were used in this process. Although the initial barium had actually been wrongly placed, a reasonable model was eventually produced in this way, having parameters close to the final values shown in Table I. The ideal formula of the compound was determined from this model to be Ba₃Nb₂₁Si₂O₄₄. After refinement of isotropic thermal parameters, using a weighting function $w = 1/\sigma_F^2$, the weighted *R*-value was 0.115 and the conventional *R* was 0.119 for the restricted data set.

Anisotropic thermal parameters were introduced for barium and niobium and the full data

				-x0120	44			
Ba(1)	Ba(2)	Nb(1)	Nb(2))	Nb(3)	Nb(4)	Nb(5) ^a	
1a	1a 2d		6g		6g	2c	1b	
$0, 0, 0 \qquad \frac{1}{3}, \frac{2}{3}, \frac{2}{50}$		$\frac{2}{21}, \frac{10}{21}, \frac{7}{100}$	$\frac{1}{7}, \frac{5}{7}, \frac{2}{1}$	<u>21</u> 00	$\frac{3}{7}, \frac{1}{7}, \frac{35}{100}$	$0, 0, \frac{21}{100}$	$0, 0, \frac{1}{2}$	
0	\$	0.1123(2)	0,1423	(2)	0.3938(2)	0	0	
0	$\frac{2}{3}$	0.4762(2)	0.6951	(2)	0.1317(2)	0	0	
0	0.4406(1)	0.0765(1)	0,2165	(1)	0.3601(1)	0.2112(1)	$\frac{1}{2}$	
1.582	2.623	0.801(52)	0.740(52)	1.757(68)	1.586	2.313	
1.582	2.623	0.814(53)	0.851(53)	1.415(62)	1.586	2.313	
0.738(52)	0.872(42)	0.248(28)	0.306(2	28)	0.383(30)	0.542(57)	1.197(145)	
0.791(31)	1.311(29)	0.391(46)	0.461(4	45)	1.094(57)	0.793(35)	1.157(91)	
0	0	0.042(35)	-0.100(3	34) -	-0.185(39)	0	0	
0	0	-0.045(35)	0.000(3	36) -	-0.132(39)	0	0	
Si(1)	O(1)) C	0(2)	O(3)		O(4)	O(5)	
2d	6g		6g	6g				
$\frac{1}{3}, \frac{2}{3}, \frac{79}{100}$	$\frac{1}{7}, \frac{5}{7},$	$0 \frac{1}{21}, \frac{1}{2}$	$\frac{5}{21}, \frac{7}{50}$	$\frac{10}{21}, \frac{8}{21}$	$\frac{7}{50}$	$\frac{5}{21}, \frac{4}{21}, \frac{14}{50}$	$\frac{2}{21}, \frac{10}{21}, \frac{14}{50}$	
1	0 1265	(14) 0.07	15(14)	0 5202	2(14)	0.2358(16)	0.0992(15)	
3 2	0 7061	(15) 0.23	67(15)	0.4112	2(14) (0.1726(15)	0.0772(10) 0.4708(14)	
0.8199(3)	-0.8199(3) 0.0080(53(5)	0.1487(5)		0.2784(5)	0 2915(5)	
0.727(107) 0.964(65) 0.91	8(162)	62) 0.633(140)		0.996(162)	0.983(159)	
O(6)	O(7) C	0(8)					
6g	6g		2d					
$\frac{10}{21}, \frac{8}{21}, \frac{21}{50}$	$\frac{1}{21}, \frac{5}{21},$	$\frac{21}{50}$ $\frac{1}{3}$,	$\frac{3}{50}$					
0.4975(18)	0.0274	16)	$\frac{1}{3}$					
0.3903(18)	0.2327	(16)	23					
0.4107(7)	0.42980	(5) 0.72	47(8)					
2.159(237) 1.282(71) 0.697(258)						
	$\begin{array}{c} \text{Ba(1)}\\ \\ 1a\\ 0, 0, 0\\ \\ 0\\ 0\\ 0\\ 1.582\\ 1.582\\ 0.738(52)\\ 0.791(31)\\ 0\\ 0\\ \hline \\ \text{Si(1)}\\ \hline \\ 2d\\ \frac{1}{3}, \frac{2}{3}, \frac{79}{100}\\ \frac{1}{3}\\ \frac{2}{3}\\ 0.8199(3)\\ 0.727(107)\\ \hline \\ 0(6)\\ \hline \\ 6g\\ \frac{10}{21}, \frac{8}{21}, \frac{21}{50}\\ 0.4975(18)\\ 0.3903(18)\\ 0.4107(7)\\ 2.159(237)\\ \hline \end{array}$	Ba(1) Ba(2) 1a 2d 0, 0, 0 $\frac{1}{3}$, $\frac{3}{250}$ 0 0.4406(1) 1.582 2.623 0.738(52) 0.872(42) 0.791(31) 1.311(29) 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 2d 6g $\frac{1}{3}$, $\frac{2}{3}$, $\frac{79}{1000}$ $\frac{1}{7}$, $\frac{5}{7}$, $\frac{1}{3}$ 0.12650 $\frac{1}{3}$ 0.12650 $\frac{1}{3}$ 0.12650 $\frac{1}{3}$ 0.707010 0.8199(3) 0.00800 0.727(107) 0.964(1) O(6) O(7)	Ba(1) Ba(2) Nb(1) 1a 2d 6g (general) 0, 0, 0 $\frac{1}{3}$, $\frac{4}{3}$, $\frac{2}{51}$ $\frac{2}{211}$, $\frac{10}{210}$, $\frac{7}{100}$ 0 $\frac{1}{3}$ 0.1123(2) 0 $\frac{1}{3}$ 0.1123(2) 0 $\frac{1}{3}$ 0.4762(2) 0 0.4406(1) 0.0765(1) 1.582 2.623 0.801(52) 1.582 2.623 0.814(53) 0.738(52) 0.872(42) 0.248(28) 0.791(31) 1.311(29) 0.391(46) 0 0 0.042(35) 0 0 -0.045(35) Si(1) O(1) C 2d 6g 1 $\frac{1}{3}$ 0.1265(14) 0.07 $\frac{1}{3}$ 0.7061(15) 0.23 0.8199(3) 0.0080(5) 0.14 0.727(107) 0.964(165) 0.91 O(6) O(7) C 6g 6g 1 121, $\frac{8}{21}$ 1 1	Ba(1) Ba(2) Nb(1) Nb(2) 1a 2d 6g (general) 6g 0, 0, 0 $\frac{1}{3}$, $\frac{4}{3}$, $\frac{2}{51}$ $\frac{2}{211}$, $\frac{10}{17}$, $\frac{7}{100}$ $\frac{1}{7}$, $\frac{5}{7}$, $\frac{1}{7}$ 0 $\frac{1}{3}$ 0.1123(2) 0.1423 0 $\frac{1}{3}$ 0.4762(2) 0.6951 0 0.4406(1) 0.0765(1) 0.2165 1.582 2.623 0.801(52) 0.740(3) 1.582 2.623 0.814(53) 0.851(3) 0.738(52) 0.872(42) 0.248(28) 0.306(2) 0.791(31) 1.311(29) 0.391(46) 0.461(4) 0 0 -0.042(35) -0.100(3) 0 0 -0.045(35) 0.000(3) 1 0.1265(14) 0.0715(14) 0.300(3) 0 0 -0.045(35) 0.000(3) 1 0.1265(14) 0.0715(14) 0.727(107) 0.8199(3) 0.0080(5) 0.1453(5) 0.727(107) 0.4975(18) 0.0274(16) $\frac{1}{3}$ $\frac{1}{3}$, $\frac{3}{3}$, $\frac{3}{50}$ 0.4903(18) 0.2327(16) <td>Ba(1) Ba(2) Nb(1) Nb(2) 1a 2d 6g (general) 6g 0, 0, 0 $\frac{1}{3}, \frac{2}{51}, \frac{2}{21}, \frac{2}{21}, \frac{1}{21}, \frac{7}{100}, \frac{7}{100}, \frac{5}{1}, \frac{5}{5}, \frac{2}{100}$ 0 0 $\frac{1}{3}$ 0.1123(2) 0.1423(2) 0 $\frac{3}{4}$ 0.4762(2) 0.6951(2) 0 0.04406(1) 0.0765(1) 0.2165(1) 1.582 2.623 0.801(52) 0.740(52) 1.582 2.623 0.814(53) 0.851(53) 0.738(52) 0.872(42) 0.248(28) 0.306(28) 0.791(31) 1.311(29) 0.391(46) 0.461(45) 0 0 -0.045(35) 0.000(36) - 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TABLE I ATOMIC PARAMETERS FOR B2-Nb- SI-O.

^a Population parameter 0.79(1).

Anisotropic thermal vibration expressed as:

$$\exp[-1/4(a^{*2}U_{11}h^2 + b^{*2}U_{22}k^2 + c^{*2}U_{33}l^2 + 2a^*b^*U_{12}hk + 2a^*c^*U_{13}hl + 2b^*c^*U_{23}kl)]$$

Errors in the last digit are given in parentheses; where no error is shown, the parameter is not independent. Isotropic thermal motion expressed as: $[\exp -8\pi^2 U \sin^2 \theta / \lambda^2]$

set was used for the final stages of refinement. Allowing population parameters to vary resulted in a significant change only for Nb(5), whose final refined occupancy factor is 0.79. The compound formula is thus Ba₃Nb_{21-x}Si₂O₄₄ $(x \simeq 0.2)$. A correction for secondary extinction was made to the 118 reflections having intensities greater than 10,000 and $\sin\theta/\lambda$ less than 0.40, using the empirical method described by Stout and Jensen (4). Anomalous dispersion was included in all structure factor calculations. Neutral atom scattering factors taken from the International Tables (6) were used throughout. The final weighted R was 0.098, the conventional R 0.124. The maximum shift to error ratio in the final cycle of refinement was 0.084. A final ΔF map showed no peak higher than one-third the height of the peak produced by deliberate omission of a single oxygen atom. Atomic parameters are listed in Table I.

Discussion

The unit cell of $Ba_3Nb_{21-x}Si_2O_{44}$ is shown in stereo in Fig. 1. It may be conveniently described in terms of the layer structure diagrammed in Fig. 2. Considering each layer simply as a closepacking of equal spheres, the stacking sequence is *ABCBCBC*. The compositions of the seven layers are: BaO_6 , O_6 (an oxygen layer with one-seventh of the oxygens missing), O_7 (a complete oxygen layer), BaO_6 , BaO_6 , O_7 , O_6 , respectively.

The BaO₆ layer found here has not, to our knowledge, been reported previously. It is, however, analogous to the KF_6 layer found in KOsF₆ and similar compounds. KOsF₆ has a structure based on a cubic stacking of approximately close-packed KF₆ layers, with one Os occupying the only available octahedral site between each pair of layers (7). Between the two



FIG. 1. Stereo view of the unit cell of $Ba_3Nb_{21-x}Si_2O_{44}$. Shaded circles are Ba, large open circles O, small open circles Nb, and small closed circles Si.



adjacent BaO₆ layers in Ba₃Nb_{21-x}Si₂O₄₄ there is, similarly, a single niobium (Nb(5)) in the only available octahedral site, although here its popu lation parameter is somewhat less than one.



FIG. 3. Niobium and silicon coordination in Ba_3Nb_{21-3} , Si_2O_{44} . See Table II for explanation of superscripts.



FIG. 4. Stereo view of Nb₆O₁₈ cluster.

Between BaO_6 and O_7 layers there are three niobiums in octahedral sites; between BaO_6 and O_6 layers there are three niobiums in sites coordinated by square pyramids of oxygens with the niobiums in the pyramid bases; and between O_6 and O_7 layers there are three niobiums in square pyramidal sites, one niobium in a normal octahedral site, and one silicon in a tetrahedral site. The coordination of the various niobiums and of the silicon is shown in Fig. 3.

The square pyramidal sites mentioned above would be octahedral except for the oxygen vacancy in the O₆ layer. The absence of an oxygen at the site is established by the lack of any peak on the final ΔF map and by the fact that, when an oxygen atom was introduced there, upon refinement it quickly acquired a thermal parameter so large ($U \simeq 55$) as to indicate that the atom was not there at all. The six niobiums in square pyramidal sites surround the vacancy in an octahedral arrangement, as shown in Fig. 4. The Nb-O distances, as shown in Table II,

TABLE II

INTERATOMIC DISTANCES AND ANGLES IN Ba3Nb21-xSi2O44

Distances:	
Ba(1)-O(1)	$2.907\pm0.014~{ m \AA}$
Ba(1)–O(2)	2.933 ± 0.009
Ba(2)-O(5)	3.018 ± 0.008
Ba(2)–O(6) (same layer)	3.039 ± 0.017
Ba(2)-O(6) (different layers)	2.953 ± 0.013
Ba(2)–O(7)	3.008 ± 0.012
Nb(1)-Nb(1)	2.793 ± 0.002
Nb(1)-Nb(2) ¹	2.840 ± 0.002
$Nb(1)-Nb(2)^{2}$	2.854 ± 0.002
Nb(1)-O(1) ¹	2.080 ± 0.012
$Nb(1)-O(1)^{2}$	2.028 ± 0.011
Nb(1)-O(1) ⁴	$\textbf{2.197} \pm \textbf{0.008}$
Nb(1)-O(2)	2.076 ± 0.011
Nb(1)-O(3)	2.115 ± 0.007
Nb(2)-Nb(2)	$\textbf{2.783} \pm \textbf{0.002}$
Nb(2)–O(2)	2.104 ± 0.010

Distances:	,
Nb(2)-O(3)	2.154 ± 0.006
Nb(2)-O(4)	2.185 ± 0.009
$Nb(2) - O(5)^{1}$	2.037 ± 0.011
$Nb(2) - O(5)^{3}$	2.114 ± 0.006
Nb(3)-O(4)	1.969 ± 0.012
Nb(3) - O(5)	2.065 ± 0.011
$Nb(3) - O(6)^{1}$	1.947 ± 0.013
$Nb(3)-O(6)^2$	1.973 ± 0.014
Nb(3)-O(7)	1.797 ± 0.007
Nb(3)–O(8)	$\textbf{2.378} \pm \textbf{0.008}$
Nb(4)O(2)	1.973 ± 0.011
Nb(4)-O(4)	1.992 ± 0.011
Nb(5)-O(7)	$\textbf{2.078} \pm \textbf{0.012}$
Si(1)-O(3)	1.622 ± 0.013
Si(1) = O(8)	1.595 ± 0.015
Angles:	
$O(1)^{1}-Nb(1)-O(1)^{4}$	97.3 ± 0.4
$O(1)^{1}-Nb(1)-O(1)^{4}$	82.1 ± 0.4
$O(1)^2 - Nb(1) - O(1)^4$	87.2 ± 0.4
$O(1)^2 - Nb(1) - O(2)^1$	86.2 ± 0.4
$O(1)^4 - Nb(1) - O(2)^1$	94.5 ± 0.4
$O(1)^{1}-Nb(1)-O(3)^{3}$	90.0 ± 0.4
$O(1)^4 - Nb(1) - O(3)^3$	87.0 ± 0.4
O(2)-Nb(1)-O(3)	86.2 ± 0.4
O(2)-Nb(2)-O(3)	91.3 ± 0.3
O(2)-Nb(2)-O(4)	79.3 ± 0.3
O(2)-Nb(2)-O(5)	90.5 ± 0.3
O(3) - Nb(2) - O(4)	87.2 ± 0.3
O(3)-Nb(2)-O(5)	87.9 ± 0.4
$O(4)^{3}-Nb(2)-O(5)^{1}$	95.2 ± 0.4
$O(4)^{3}-Nb(2)-O(5)^{3}$	88.6 ± 0.3
O(4) - Nb(3) - O(5)	88.0 ± 0.4
O(4) - Nb(3) - O(6)	96.1 ± 0.6
O(4) - Nb(3) - O(7)	101.8 ± 0.4
O(4) - Nb(3) - O(8)	84.3 ± 0.4
O(5)-Nb(3)-O(6)	82.0 ± 0.3
O(5)–Nb(3)–O(7)	94.5 ± 0.4
O(5) - Nb(3) - O(8)	84.2 ± 0.2
O(6) - Nb(3) - O(6)	88.8 ± 0.5
$O(6)^{1}-Nb(3)-O(7)^{3}$	101.6 ± 0.5
$O(6)^2 - Nb(3) - O(7)^3$	95.4 ± 0.4
O(6) - Nb(3) - O(8)	78.7 ± 0.4
O(2)-Nb(4)-O(2)	91.7 ± 0.5
$O(2)^{1}-Nb(4)-O(4)^{1}$	87.3 ± 0.4
$O(2)^{1}-Nb(4)-O(4)^{2}$	89.8 ± 0.4
O(4)-Nb(4)-O(4)	91.2 ± 0.4
O(7)-Nb(5)-O(7) (O's in same layer)	$91.1 \pm 0.5^{\circ}$
O(7)-Nb(5)-O(7) (O's in different layers)	88.9 ± 0.5
O(3)-Si(1)-O(3)	110.0 ± 0.5
O(3)–Si(1)–O(8)	109.0 ± 0.4

TABLE II—continued

Where necessary, symmetry-related atoms are distinguished by superscripts referring to the list of general position coordinates for space group $P\overline{3}$ (No. 147) in the International Table (11).

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TABLE III

Electrostatic Bond Strengths to Oxygen in $Ba_3Nb_{21-x}Si_2O_{44}$

Assuming all niobium to be in average oxidation state (+3.56):

O(1): 1 Ba at 1/6 + 3 Nb at 0.71	= 2.30
O(2): 1 Ba at 1/6 + 2 Nb at 0.71 + 1 Nb at 0.5	9 = 2.18
O(3): 1 Si at 1 + 2 Nb at 0.71	= 2.42
O(4): 1 Nb at 0.71 + 2 Nb at 0.59	= 1.89
O(5): 1 Ba at 1/6 + 2 Nb at 0.71 + 1 Nb at 0.5	9 = 2.18
O(6): 2 Ba at 1/6 + 2 Nb at 0.59	= 1.51
O(7): 1 Ba at 1/6 + 1.8 Nb at 0.59	= 1.23
O(8): 1 Si at 1 + 3 Nb at 0.59	= 2.77

Assuming Nb(1) and Nb(2) to be in +2.5 oxidation state and Nb(3), Nb(4), and Nb(5) to be Nb^v:

range from 2.03 Å to 2.20 Å, well within the normal range for niobium oxides, but the Nb–Nb distances, averaging 2.81 Å, are extremely short.

In the ion $Nb_6O_{19}^{8-}$, for instance, the arrangement of niobium and oxygen is the same as in the present Nb₆O₁₈ grouping except that an additional oxygen is present in the center of the group. The shortest Nb-Nb distance there, however, is about 3.2Å (8). In that case, niobium is present as Nb^v. In $Ba_3Nb_{21-x}Si_2O_{44}$, however, the average niobium oxidation state is +3.56, assuming x = 0.21. Moreover, Nb(3), Nb(4), and Nb(5) are found in the centers of simple NbO₆ octahedra joined by corner- and edgesharing and probably have oxidation numbers not much below +5. Hence Nb(1) and Nb(2), which make up the octahedral cluster, probably have oxidation states in the neighborhood of +2.5.

Thus a more appropriate analogy than the Nb₆O⁸₁₉ group is found in the Nb₆Cl²⁺₁₂ ion, which consists of an octahedron of niobiums with chlorines placed some distance out from the midpoints of the edges. The niobium oxidation state is $+2\frac{1}{3}$, the average Nb–Nb distance 2.9 Å, and the shortest distance 2.85 Å, quite comparable to the values in Ba₃Nb_{21-x}Si₂O₄₄ (9). An even more similar Nb–Nb distance is found

in NbO₂, in which edge-sharing NbO₆ octahedra in a rutile-type chain are distorted so as to bring pairs of niobiums into metal-metal bonding range. The Nb-Nb distances along the chain are alternately 2.80 Å and 3.20 Å (10). The bonding distance of 2.80 Å is virtually identical to the 2.81 Å Nb-Nb distance found in the present compound.

Nb(1) and Nb(2), then, form an octahedral cluster of niobiums held together by metalmetal bonding, each niobium being coordinated to four others. The coordination sphere of each niobium consists of five oxygens in a square pyramidal arrangement, with the niobium in the base of the pyramid. Nb₆O₁₈ clusters can be identified in which the Nb₆ octahedron is surrounded by a cuboctahedron of oxygen atoms capped by oxygen atoms on the square faces. Alternatively, the oxygen array may be described as an octahedron with oxygens also at the centers, of the edges. The remaining niobiums in the structure are in octahedral sites, having Nb-O distances of 1.95 to 2.08 Å, with the exception of the Nb(3)-O(8) distance of 2.38 Å. The extreme length of this bond relative to the normal Nb-O distance is due to the very poor local charge balance around O(8) shown in Table III. Electrostatic bond strengths are given first

BARIUM NIOBIUM SILICON OXIDE

TABLE IV

Deserved and Calculated Structure Factors. Columns Are l, $10|F_0|$, $10|F_c|$. Unobserved Reflections Are MARKED WITH ASTERISKS.

	2 349 2 349 10 10 2 349 2 349 10 10 2 349 2 349 10 10 10 349 349 349 10 10 10 349 349 349 349 10 10 10 349	
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5.000 5.000 1.1266.250 1.1266.250 2.785.760 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1535 4.1264.1555 4.1264.15577 4.1264.15577 4.1264.15577 4.1264.155777 4.1264.15577777	10 10 10 10 10 2224 10 10 10 20 247 30 20 20 10 20 247 30 20 20 20 20 20 20 247 405 20 2	• • • • • • • • • • • • • • • • • • •	13 600 -344 14 927 631 14 927 631 15 927 631 15 927 631 16 927 631 17 927 73 18 927 631 19 927 631 10 927	0 2744 2113 2 171 -33 2 2407 -233 3 2407 -233 4 5 344 367 5 443 -235 6 443 -235 6 443 -235 7 5 443 -435 4 437 -235 13 410 -235 14 431 -235 14 431 -235 14 431 -235 14 431 -235 15 445 14 431 -235 15 445 15 445 15 445 15 445 15 445 15 445 15 445 15	12 314* 374* 13 344* -756* 14 345* -756* 15 360* -730* 16 321* -131* 16 321* -131* 17 221* -137* 18 321* -137* 19 335* -17* 10 -32* -17* 10 -32* -17* 10 -32* -17* 11 -32* -17* 12 -32* -17* 13 -32* -17* 14 -35* -32* 15 -32* -17* 16 -32* -17* 17 -32* -10* 13 -32* -10* 14 -32* -10* 15 -32* -10* 16 -32* -10* 17 -32* -10* 18 -32*	1 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0		16 602 -360 17 2006 104 10 602 208 20 314* 222 21 244* -322 21 244* -322 21 244* -322 22 24* -322 23 7* -7 400* 315 -132 315 -132 201* 315 -132 204* 315 -132 204* 315 -132 204* 315 -132 204* 315 -132* 104* 315 -132* 104* 315 -132* 104* 316 -305* 304* 317 205* -104* 11 275* -104* 12 204* -105* 13 305* -205* 14 205* 104* 15 20	17 1000 18 1000 19 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 1000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 10000 10 100000	11 407 -242 22 2407 -251 23 2407 -136 24 251 251 25 257 -136 25 257 -136 26 257 -136 27 251 511 28 257 -136 29 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 -737 20 207 207 <	4 338 4 338 7 6 2607 6 2607 7 6 2607 7 7 6 2607 7 8 2607 7 8 2607 10 2607 1
1 505 502 2 1212 1191 3 587 -559 4 263 205 5 186 -215 4 370 576	26 2344 201 27 201 108 28 411 395 29 313 334	32 332 354 33 316 -335 61-5+L 1 852 865	19 230# 14 20 543 567 21 248 149 22 1192 1320 23 547 234 24 238# 85	7 230# 127 8 306 -310 9 518 -582 10 203# 22 11 277# -10 12 392 -478	10 309* 268 7,-7.L 1 189 -203 2 1712 -1684	25 350 -450 26 258 -321 27 465 593 28 278 154 29 506 604 30 728+ 219	16 239 205 17 255 83 18 213 122 19 279 215 20 260 -99 21 312 299	• 72 983 7 2459 72 8 326 -124 9 403 -366 10 3209 -82 11 2979 47	1 169 195 2 208 155 3 359 -277 83.L	29 365 467 8+2+L 8 246 209	2 204+ 5 250+ 4 244+ 5 254+ 6 219+

TABLE IV—continued

BARIUM NIOBIUM SILICON OXIDE

	8+8+L	20 266* 274	92.1	27 237* 17	9.4.1	10 346 290	11 205 298	10+-1+L	6 195* -12	10 317 205	125.L	147.L
,	278 -195	21 256* **	1 151* -72	28 379 238	0 657 481	11 194 -134 12 282 235	12 301 -246	1 984 822	9 350 313 10 711 593	1110.L	1 226* -157	1 744* 108
8	236+ 14	23 246* 67	2 359 -271	9#1#L	1 185* -35	13 246* -51	14 649 677	2 477 462	11 183 -197		2 228* 103	2 212 60
	290* 156	24 457 475	3 992 -862	0 413 340	3 599 576	15 350+ 371	15 200* 188	3 470 481	13 235 38	2 188* -67	3 1063 943	3 196• ~2
ĩ	330. 121	26 222 196	5 1492 1305	1 199 -248	4 209 -131	16 442 362	17 283 369	5 667 552	14 207 212	3 189* -130	12+-4+L	14+-6+L
		27 253* -48	6 324 -285	2 813 -679	5 235 71	17 274 44	18 276 -265	6 1214 1058	15 218 -60	11		
	99.L	28 220* +21	9 172 139	3 314 -235	7 400 -339	19 478 289	20 330 329	7 406 298	17 270 -325	11,-9,1	2 437 290	2 217 244
1	160+ +63	30 208* -65	10 1562 1374	5 883 814	9 175 -77	20 251 125	21 268+ -43	10 291 141	18 296 184	1 173 -104	3 508 -299	3 318 276
Z	459 444		11 628 -557	6 151 154	10 219* -136	Z1 252* 118	22 269 -114	11 359 -279	19 624 556	2 414 367		
4	529 512	9,-011	13 116 210	7 889 846	12 301 264	23 251+ -233	24 252* -95	13 1270 -1114	21 313* 140	5 872 854	124-946	141-212
5	633 743	1 445 406	14 271 232	10 1245 1109	13 267* 90	24 238 134	25 301 359	14 274* 232	22 266* 104	11+-8+L	209* -76	1 211* -41
6	436 -517	2 172 -143	15 2174 26	11 230 -181	14 583 496	25 230* 53	26 226 -142	15 1095 920	23 314 201	1 209# 79	2 180 86	2 254* 178
á	167* -136	4 460 467	17 727 -718	13 204# 58	16 316 197	10 247- 147	28 419 426	17 314 300	10.3.L	2 376 402	2 200- 42	200- 70
9	249 116	5 533 538	18 450 394	14 348 364	17 369 285	109.4		18 220 74		3 432 -358	12+*2+L	14+-4+L
c 1	253 49	6 502 -591	20 2758 -1003	15 260 -292	18 382 +280	1 445 421	10,-6,L	19 520 417	0 255* -285	11-7-1	1 357 320	1 480
ż	245 -153	8 476 -503	21 461 449	17 352 -270	20 272* -52	2 172* -117	1 1310 ~1202	21 287* 145	2 223 121		2 360 -326	2 236 129
3	506 613	9 540 668	22 262* 57	18 276 81	21 312 -148	3 166* 77	2 458 -427	22 746 -701	3 371 -292	1 313 292	3 189* -9	3 1027 694
6	319 316	10 410 525	23 3004 130	14 102 654	14 193* 80	5 709 701	3 1244 90	23 301+ -253	5 2124 19	3 201+ 6	1310.6	154-104
6	489 413	12 159 35	25 402 -340	21 305 44	9.5.L	6 581 607	105.L		6 242* -171			
7	565 +525	13 544 621	26 253 -240	22 232 310	0 380 336	7 483 -544	1 2005 4	10,0+L	7 169 238	11+-6+L	1 591 461	1 179 119
9	284 254	15 227 148	28 2279 74	23 405 557	1 200 133	9 420 514	2 577 491	0 151 -3	10 340 211	1 453 472	3 309 261	3 256* -3
٥	214 -329	16 279* -175	29 236* -26	25 244+ -149	2 193* 39	10 Z28* -248	3 1016 -959	1 183* 65	11 281* 69	2 172 41		
;	249* -25	17 298* -215 18 269 306	30 265* 25	26 346 -255	3 162 8	11 372 -433	10.00	2 173 227	12 225* 159	3 167* 65	13,-0,1	159.6
3	322 185	19 254* 2	9+-1+L		5 282 249	13 224 18	101 -12	4 161 90	14 422 -449	11+-5+L	1 239• 133	1 183 187
5	299 375	20 322 -359		9,2,1	6 263* 127	14 444 473	1 237 197	5 470 -426	15 412 -366		2 433 337	2 266 -198
2	165 -106	21 696 -/66	2 179 -21	5 533 444	9 221 16	15 437 399	2 535 -407	6 213 179	16 264 45	1 186 192	3 210* 165	3 498 -394
8	2490 72	23 603 563	3 345 -339	1 1259 1098	10 229* 29	17 270 334		9 355 268	18 279+ 95	3 191 -58	138.L	15+-8+1
	0	24 271 214	4 613 494	2 702 689	11 253 -137	18 711 743	10,-3,L	10 223 218	19 327 127			
	**-0*C	26 231* -30	6 212 131	4 439 -452	13 328+ 153	20 331+ 260	1 947 752	12 401 225	20 345* -61	114-445	2 350 237	2 2000 162
1	384 359	27 659 587	7 157 165	5 461 -445	14 406 298	21 611 -557	2 179 17	13 242 287		1 398 318	3 891 826	3 372 329
2	1217 -1106	26 774 810	9 439 347	6 395 371	15 300+ 168	22 272* 78	3 303 -275	14 2210 -86	10.4.L	2 343 -308	10.00	
	519 528	30 309 -309	11 255 -78	9 439 -399	17 2924 49	24 2424 -96	5 244 -264	16 237 -78	0 849 687	3 877 -934	131-740	13/.0
5	147 ~55		12 188* 114	10 521 410	18 285* 194	25 318 -363	6 540 476	17 249 104	1 446 425	11,-3.L	1 159 -141	1 676 -534
2	183 18	9,-5,1	13 351 327	11 760 799	19 355* 302	26 191 -232	7 211 -38	18 279* -19	2 312 -232	1 176 101	2 422 -374	2 348 -305
8	1990 -55	1 1145 1072	15 326 -307	13 213 61	9,6.6	21 234 -12	10 359 253	20 400 246	4 371 265	2 226 170	5 6 - 557	244- 102
2	790 845	2 376 360	16 262* -165	14 476 414		10,-8,L	11 254 -186	21 304+ 116	5 231+ 18	3 778 558	136.L	156.L
3	629 752	3 506 485	17 269* -63	15 999 843	0 205* -50	1 259 244	12 183* 117	22 304 224	7 430 -326	112.	1 1894 -40	1 299 202
ż	359 418	9,-4,L	19 280 185	17 338 -202	2 323 239	2 229 -183	14 201* 87	24 391 251	9 528 438		2 219* . 8	2 216+ 20
3	251• -91		20 274* 132	18 657 -610	3 245* 54	3 830 -823	15 863 689	25 248* 67	10 318 182	1 899 -787	3 210 -35	3 320 -317
ŝ	262# 116	2 761 732	21 265* 130	20 320 288	5 316 278	5 136 -105	10 211 -275	26 200 150	12 250* 120	3 359 286	135.6	151-516
6	730 -733	3 161* 54	23 336* 97	21 410 -460	6 278 -141	6 727 761	18 478 410	10.1.L	13 275* 118			
:	231 -140	0.001	24 225 275	22 269 -429	7 490 439	7 593 637	19 266* -67		15 3210 225	11,-1,0	2 236* 184	1 427 -21
ą.	254 30	41-576	26 246* 82	24 674 628	10 226* -185	9 601 579	21 306* -141	1 705 -564	16 279+ -232	1 206* 131	3 430 -361	3 455* 92
0	255+ 125	1 1272 -970	27 243* -74	25 490 611	11 334 -253	10 462 490	22 313* -116	2 287 195	17 289* -152	2 200 164		
2	251 60	2 577 423	28 209 -323	26 275 275	13 311 148	12 275 200	25 2624 -137	3 669 756	10 310 177	5 050 - 707	134.1	16,-10,0
3	679 625	4 911 808		9+3+6	14 283 -149	13 354 -441	26 306* 184	5 407 353	10.5.L	1210.L	1 202 -212	1 230* ~45
4	291 247	5 639 537	9.0.1		15 315* -22	14 217* -45	27 256* -61	6 470 358	0 204* 41	1 267 214	2 512 -310	2 266• 76
6	282 238	7 262 222	0 847 753	1 1814 -38	10 201 210	16 223 -122	10 274- 98	9 230* 63	1 204+ -30	2 249* -148	,	3 400 401
7	198 -169	9 191* ~34	1 611 578	2 169 96	9.7.L	17 413 -558	102.1	10 805 -672	2 303 -219	3 215* -89	13,-3,L	16+-9+L
8	2264 -90	10 1375 -1247	2 218 -226	3 187* 148	0 225 152	19 264 - 3	1 278 258	11 252 167	4 247 51	129.4	1 1168 932	1 7760 69
		12 145 -167	4 737 652	5 235 176	1 216* 17	20 819 796	2 221 139	13 299 -197	5 417 322		2 341 299	2 238 9
	9,=7,L	13 241 -206	5 191* -48	6 186* 93	2 540 -555	21 361 287	3 319 -197	14 307 269	6 200 195 7 100 - 45	1 199* -27	3 270* -114	3 205+ -79
	636 615	14 307 318	6 154 97 7 156	9 241 144	4 292 166	23 473 423	5 671 536	15 446 -370	9 343 311	3 538 -379	14,-10,4	168.4
ż	313 300	16 875 796	9 884 810	10 235+ -116	5 236 172	24 261 215	6 717 -576	17 667 627	10 550 447			
3	325 -303	17 1096 1029	10 341 287	11 286 -106	6 249 73 7 2380 -118	25 247* 228	7 638 -589	18 334 210	11 290 ~82	128.L	1 200* 83	1 206* 38
5	432 -489	19 278* 102	12 272 317	13 227 -20	9 520 544	27 341 -274	10 589 457	20 253 160	13 243+ +30	1 180* -89	3 378 -288	3 241* -21
6	336 3AQ	20 258* 124	13 206* -77	14 231 -132	10 424 402	28 2210 -57	11 316 262	21 234 -95	14 372 80	2 775 -651	14	
á	162 80 301 317	21 275 - 29	14 572 466	15 260 48	++ 200140	107.1	12 216* -140	22 339 337	20010	5 141 -158	141-41	16/.
9	143 84	23 406 -322	16 262* -186	17 311+ 180	10,-10,L		14 1377 1234	25 2474 119	10.6.L	127.L	1 190 -23	1 204+ 176
2	442 418	24 732 -814	17 224 -172	18 496 456	1 220 327	1 202 123	15 262 135		0 929 740	1 1165 1020	2 576 -410	2 294 136
ż	1910 160	26 201+ -157	19 327 -291	20 201+ 119	2-386 328	3 833 745	17 268 -309	10.2.1	1 585 474	2 222 20Z	5 234 -74	2 219- 14
3	354 360	27 233* 144	20 311* 147	21 242 127	3 201 230	* 224 -125	18 331 275	0 200 154	2 490 450	3 232* -171	14,-8.L	16+-6+6
5	482 577	28 238 195	21 3364 -110	22 259* 9	+ 197 155 5 586 480	> 729 712 6 726 724	19 302 64	1 193 -94	- 201• -41 4 348 -344	126.1	1 517 487	1 2094 10
6	2390 -109	30 380 513	23 645 586	24 274 -175	6 183 144	7 311 -212	21 496 -496	3 427 - 390	5 185 -156		2 459 365	2 239+ 93
7	300 -249		24 304* 117		7 188 105	.8 207 158	22 373+ 60	4 211 171	6 208* 46	1 275 -267	3 257 -253	3 293 ~89
н 9	277 88		25 261* -43		8 236 -110		23 417 358	5 774 620	9 264 -284	3 229# 66		

assuming that all Nb have the average oxidation state of 3.56, and then supposing Nb(1) and Nb(2) to be +2.5 and Nb(3), Nb(4), and Nb(5) to be +5. The second supposition gives more reasonable bond strengths in five out of eight cases; it makes the situation worse only for O(8), and the lengthening of Nb–O(8) distances may be sufficient compensation for the abnormal bond strength calculated to that atom even under the second distribution of oxidation states. This distribution is therefore presumed to be more nearly correct than the original assumption of average oxidation states.

The SiO₄ tetrahedron is quite regular, with Si–O distances ranging from 1.60 to 1.62 Å. Ba(1) has a very regular 12–fold coordination, with Ba–O distances 2.91 to 2.93 Å. The coordination about Ba(2) is somewhat distorted with Ba–O distances ranging from 2.95 to 3.04 Å; the barium is displaced out of its layer towards the oxygens of the adjoining BaO₆ layer, thus helping to hold the two layers together by electrostatic attraction.

Observed and calculated structure factors are shown in Table IV. The relatively poor agreement is attributed to the instrumental instability. However, convergence was attained.

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